# RESEARCH PAPER

# A Practical Method to Predict Physical Stability of Amorphous Solid Dispersions

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# ABSTRACT

**Purpose** To predict the crystallization time of amorphous solid dispersions by controlling the combined effect of temperature and moisture content.

**Methods** The authors exposed amorphous samples of spraydried API and Hydroxypropylmethylcellulose Phtalate to various temperature and humidity conditions below and above the glass transition temperature (Tg) until crystallization of the API was observed. The crystallization of API was detected by XRPD, while the  $T_g$  and the water absorption by the amorphous dispersion are quantified by mDSC and water sorption analysis.

**Results** Extrapolation of the data obtained at a temperature above  $T_g$  to conditions below  $T_g$  gives only a qualitative trend. By contrast, in conditions below  $T_g$  the logarithm of onset of crystallization time was shown to vary linearly with the  $T_g/T$  ratio. A statistical analysis shows that the data obtained in the highest temperature/humidity conditions, for which the onset of crystallization is below 3 months, can be extrapolated over 15 months.

**Conclusions** The proposed methodology can be used as a stress program to predict long-term stability from a relatively short observation period and to design appropriate temperature and humidity conditions for long-term storage to prevent crystallization.

**KEY WORDS** amorphous · crystallization onset · glass transition temperature · moisture · physical stability

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## INTRODUCTION

The proportion of drugs exhibiting a poor bioavailability due to their poor water solubility has significantly increased in the last years (1). One of the most successful ways to solve this problem is to improve the drug release of such pharmaceutical solids by developing an amorphous form, generally as a solid dispersion with polymeric carriers (2–5). The high energy state of the amorphous form confers to it higher solubility and enhanced bioavailability through improved dissolution. Consequently, the use of amorphous phases has been the subject of very intensive investigations in the pharmaceutical field (6–11).

One of the main challenges of these solid amorphous forms is the physical instability associated with the high energy state, and the difficulty to forecast potential instability. How can one predict rapidly, during the evaluation of different candidate formulations, the long term stability (for instance over 2 years) of these metastable products? The extrapolation of accelerated conditions, classically used for chemical degradation (ICH Q1E guideline)-which claims that data for 6 months at 40°C 75%RH is predictive for 24 months at 25°C/60% RH- cannot be applied here. Hence real time stability studies, both time-consuming and costly, are necessary to anticipate physical stability. The aim of this article is to propose a methodology that would enable the prediction of long term physical stability from appropriate accelerated studies.

Our objective is hence very practical, but in order to attain it, this study must be based on the theoretical equations describing the mechanisms which underlie the instability of amorphous systems. Numerous publications have shown that the molecular mobility of the amorphous phase is responsible for chemical (6,12) and physical instabilities (9,13-17). It is commonly known that the molecular

mobility of an amorphous compound is due to two main relaxations (18,19), called  $\alpha$ - and  $\beta$ - relaxations. The  $\beta$ relaxations are linked to local molecular movements and have an Arrhenius behaviour over the whole range of temperature. The  $\alpha$ -relaxations are linked to cooperative molecular motion. For these movements, the temperature dependence of the relaxation time is described by the Adam Gibbs model, with three different regimes, described in Fig. 1. Above  $T_{\epsilon}$  (Cross over temperature where  $\alpha$  and  $\beta$ relaxation times are similar), the  $\alpha$  -relaxation time follows Arrhenius behaviour. For temperatures Between  $T_c$  and  $T_{g}$ , the  $\alpha$  relaxation time exhibits a non linear temperature behavior that follows the Vogel-Tammann-Fulcher (VTF) law (see theoretical interpretation part). Finally below  $T_{\alpha}$ the molecular structure of the amorphous phase is frozen and the relaxation time behaves again according to the Arrhenius law (18).

Among the numerous attempts to correlate molecular mobility of the amorphous solids with physical instability, some work can be highlighted here. Alié and co-workers (14) attributed the crystallization rate of a pure amorphous drug to  $\beta$  relaxations in the temperature range above  $T_{g}$ , suggesting that molecular rearrangements were the rate limiting step. More frequently the crystallization times are attributed to  $\alpha$ -relaxations, as diffusion seems to be the main limitation for the crystallization kinetics. Both cases may however exist depending on the molecular properties and the rate limiting step.

For phenobarbital, nifedipine and their dispersions with PVP, Aso and collaborators (17,20) demonstrated a strong correlation between the  $\alpha$ -relaxation times, measured by DSC and solid state NMR, and the crystallization rates, above and below  $T_g$ . Bhugra and co-workers (13,21) evaluated the relaxation times of amorphous sucrose by dynamic relaxation spectroscopy, isothermal microcalorimetry and modulated DSC, and showed that above  $T_g$  a strong coupling exists

**Fig. 1**  $\alpha$  and  $\beta$  relaxation times represented as a function of  $T_g/T$  (adaptated from 18).

between these measurements and the crystallization onset. They then used this correlation in order to predict "semiquantitatively" the crystallization onset below  $T_{r}$ .

These studies evaluated the impact of the temperature, but another parameter also plays a very important role: the humidity. Indeed, the humidity is responsible for decreasing the  $T_{\sigma}$  of amorphous solid samples due to the plasticizing effect of water, and consequently favors the crystallization phenomenon. Interestingly, Miyazaki et al. (22) showed how the humidity and the temperature can be coupled in a unique parameter,  $T_g/T$ . In the  $T > T_g$  region, they found that the crystallization rate of amorphous nifedipine has an Arrhenius-like behaviour when expressed as a function of  $T_{a}/T$ . Similar results were obtained by Schmitt *et al.* (23) on spray dried amorphous lamotrigine mesylate. Indeed, when exposing the drug to moisture, a parallel decrease of  $T_{\rho}$  and crystallization onset was observed by DSC (hermetic pans). Based on their published data, we calculated that the ratio  $T_{\rho}/T_{onset}$  (in K) is approximately constant (0.84–0.87) (Table I). These data, represented in Fig. 2, suggest that temperature rescaling by  $T_g$  may reveal interesting properties considering physical stability.

As it has been demonstrated in the literature, it is possible to utilize the combined effect of temperature and humidity in order to simply correlate the crystallization kinetic with a unique parameter  $T_g/T$ . In this study, we propose to use this idea in order to extrapolate physical stability at a long time period from data obtained at low temperatures ( $\leq 60^{\circ}$ C) and high humidities. For this purpose, we used a Sanofi drug substance called SAR. SAR is a very insoluble BCS class II compound, formulated as a spray dried amorphous solid dispersion in HPMCP (20% SAR/80% HPMCP w/w). We propose to expose this sample to different temperatures and humidities and to characterize the crystallization onset for different  $T_g/T$  values. For the long term sub  $T_g$  studies, we used the methodology developed by Miyazaki (22), which



**Table I**  $T_{g}$ , Temperature of Onset of Crystallization for Lamotrigine Mesylate from Schmitt et *al.* (23) as a Function of Moisture Content - The  $T_g/T_{onset}$  Ration was Calculated by the Authors

Moisture%	T <sub>g</sub> (°C)	T <sub>onset</sub> crystallisation (°C)	T <sub>g</sub> /T <sub>onset</sub> (Kelvin values)
0.71	98.6	155.6	0.87
0.89	90.9	155.2	0.85
1.25	82.7	152.7	0.84
1.74	68.7	123.7	0.86
2.28	58.6	113.3	0.86
2.75	49.8	105	0.85
3.56	40.8	93.3	0.86
3.83	38.6	89.3	0.86
4.09	31.2	85.3	0.85

modulates  $T_g$  by exposing the product to various relative humidity values.

As has been reported by several authors (13,14,21), we characterize the crystallization by determining the onset of crystallization time " $\tau_{oc}$ ". The detection of the crystallinity of the drug substance is a critical experimental point in this study. Indeed, from a practical point of view, it is the end of shelf life: until crystals are experimentally detected, the product will be considered as still amorphous and compliant. Nucleation however probably occurs slightly before. In numerous papers the crystallization rate was measured by DSC or by optical microscopy (24,25) for pure amorphous drugs or solid dispersions with a maximum of 20-25% of polymer (17,20,22,26). XRPD was used by Yoshioka et al. (27) to quantify precisely the degree of crystallinity of initial pure amorphous drug. Onset times of crystallization have been measured at  $T > T_g$  either by DSC (13); or by dielectric spectroscopy isothermally (14). Onset times for  $T < T_{\sigma}$  have been measured by polarized microscopy (13) In this work, a simple and straightforward XRPD method was developed



**Fig. 2** Spray dried amorphous amorphous lamotrigine mesylate  $T_{g'}$  crystallization onset and  $T_g/T_{onset}$  ratio vs. moisture content adapted from Schmitt et *al.* (23).

to measure the crystallization onset for solid dispersions at low drug loads (20%).

The different steps of the present study are the following:

- To establish the impact of temperature and humidity on physical stability over a long time period (in the  $T < T_g$  domain)
- To determine if long term (15 month) stability could be predicted on the basis of 3 month "stress condition" data (in the *T* < *T<sub>e</sub>* domain)
- To perform some measurements for T > T<sub>g</sub> in order to determine if there is a reasonable chance to predict sub T<sub>g</sub> physical stability based on crystallization kinetics at T>T<sub>g</sub>

# MATERIALS AND METHODS

#### Materials

SAR is a drug compound developed by Sanofi-Aventis. HPMCP HP55 was supplied by Shin-Etsu Chemicals Co. (Tokyo, Japan). For the spray-drying process, EtOH and methylene chloride were purchased from Solvay (Brussels, Belgium). Silica gel and NaCl were obtained from VWR International, LLC (Radmon, Pennsylvania, USA). K<sub>2</sub>SO<sub>4</sub>, NaBr and (NH<sub>4</sub>)<sub>2</sub>SO<sub>4</sub> are purchased from Thermo Fisher Scientific Inc. (Waltman Massachusetts,USA).

#### Preparation of the Sample by Spray-Drying

SAR was dissolved with HPMCP HP55 in the solvent mixture  $CH_2Cl_2/EtOH 95\%/5\%$ . The ratio SAR/ HPMCP was 20%/80% and the total solid concentration was 6.2% (all percentages are given in weight ratios). The final solution was atomized with a large pilot-scale loop spray-drier supplied by the Drytec company (diameter: 1 m, length: 3 m, gas flow approx. 500 kg/h). Inlet and outlet temperatures were fixed at 95°C and 50°C. A bifluid nozzle was used. After spray drying, the residual solvents (about 3%) were fully removed to well below ICH specifications by drying under vacuum (P~50 mbar).

#### Storage of Samples in Stressed Conditions for $T < T_{e}$

The measurements performed in the low temperature domain ( $T < T_g$ ) represent the most important part of the experimental data of this study. The samples were conditioned at fixed temperature and humidities by placing them over saturated salt solutions at 60, 75, 80 and 100%RH, or over silica gel (0%RH) in closed glass desiccators either at room temperature (23°C) or in ovens at 40, 50, 55, 60 and 96°C. The storage conditions are summarized in Table II.

 Table II
 Temperature and RH of Storage, Corresponding Salts Used To Control Humidity

Storage T (°C)	Storage RH (%)	Salt
96	0	Silical gel
60	80	(NH4)2SO4
55	80	(NH4)2SO4
50	100	K <sub>2</sub> SO <sub>4</sub>
50	80	$(NH_4)_2SO_4$
50	75	NaCl
40	100	K <sub>2</sub> SO <sub>4</sub>
40	80	$(NH_4)_2SO_4$
40	60	NaBr
23	100	K <sub>2</sub> SO <sub>4</sub>
23	80	$(NH_4)_2SO_4$
23	75	NaCl

The samples were removed and analyzed by XRPD at regular intervals of time. Analyses were carried out every 2 or 3 days for up to 20 days, followed by every week for 10 weeks, and then once every month for evaluations longer than 10 weeks. Except for one sample stored at 23°C and 100%RH, the analyses were discontinued upon crystallization of the sample.

For studies carried out in the high temperature domain  $(T > T_g)$ , only the temperature was varied, the humidity being fixed to 0%RH for practical reasons (see the section on the T-XRPD analysis).

#### Modulated DSC

A MDSC Q200 supplied by TA Instruments (New Castle, Delaware USA) was used for  $T_g$  measurements. Samples from 2 to 10 mg were subjected to a modulated heating program from room temperature to 250°C, at a heating rate of 3°C/min, a period of 60s and amplitude of 1°C,. Depending on the requirements of the experiment (dry or humid conditions) hermetically or non-hermetically sealed pans were used.  $T_g$  was measured on the "reversing" heat flow signal, using the software Universal Analysis from TA Instruments. The  $T_g$  was calculated as the mid-point at the half height of the glass transition.

#### **XRPD** Analysis

A XRPD diffractometer, running at ambient temperature, was used to assess the crystallinity of the samples stored at fixed Tand humidity conditions ( $T < T_g$ ). Analyses were performed using a D8 Advance diffractometer from Bruker AXS (Madison, Wisconsin USA), with a Bragg-Brentano focusing geometry ( $\theta$ - $\theta$ ) in the reflection mode. The diffractometer is configurated with a LinxEye detector. A tube with an anticathode of copper running at 40 kV and 40 mA was used. Two lines are typically emitted: Cu*K*a1 ( $\lambda$ =1.54060 Å) and Cu*K*a2 ( $\lambda$ =1.54439 Å). The incident radiation was filtered to eliminate the K<sub>β</sub> radiation. The parameters employed were as follows: scan range from 5 to 30 °2 $\theta$ , with a step of 0.02° 2 $\theta$  and 3 s of counting per step.

The onset time of crystallization  $(t_c)$  was defined by the appearance of more than one peak in the SAR diffractogram, that is, the time when the first SAR crystals are detected by XRPD. The limit of detection of crystallinity this method has been estimated to be 3% for SAR (unpublished data).

By convention, the crystallization onset  $(t_{oc})$  was taken as the mean time between the last time where no crystallinity was detected  $(t_{nc})$  and the first time where crystallinity was detected  $(t_{c})$ :  $t_{oc} = (t_{nc} + t_c)/2$ .

# Variable T-XRPD Analysis

A XRPD diffractometer with a temperature chamber was used to assess the crystallinity of the samples exposed to temperatures above  $T_{\rho}$  and 0%RH. Tests were carried out with a Bruker D5000 diffractometer equipped with the Bragg- Brentano parafocusing  $(\theta - \theta)$  geometry and an Anton-Paar TTK450 temperature chamber flushed by dry nitrogen. The powder was deposited in a concave stainless steel sample holder. A sealed cobalt anode X-ray tube running at 40 kV and 30 mA was used on this diffractometer. Two lines are typically emitted:  $CoK\alpha 1$  $(\lambda = 1.7890 \text{ Å})$  and CoKa2  $(\lambda = 1.7929 \text{ Å})$ . An Iron  $\beta$ -filter, placed between the detector and specimen, does not altogether eliminate  $\operatorname{Co}\mathcal{K}\beta$  ( $\lambda$ =1.6208 Å) radiation, which still contributes about 1% of the diffracted beam at the detector (manufacturer's data). The beam is sighted using Soller slits, to improve its parallelism. Variable divergence slits keep the illumination area of the sample constant. A 1 mm collimator limits diffusion between the tube and the sample. A Braun 50-M multichannel linear detector completes the setup.

Temperature was allowed to increase at a rate of  $0.05^{\circ}$ C/sec. The following temperatures were used: 115, 116, 126 and 136°C. Diffractograms were recorded between 5 and 35° 2 $\theta$ , with a step of  $0.02^{\circ} 2\theta$  and 1 s of counting per step. The time of counting was reduced in comparison to standard XRPD, at room temperature, because the time of crystallization is very low for elevated temperatures (example: around 90 mn for 136°C) and rapid scans are hence necessary. Data were acquired in the isotherm mode when the requested temperature was reached.

# Water Sorption Cycle

The sorption-desorption isotherms were recorded by a VTI SGA100 instrument. The SGA100 measures the uptake and loss of vapour gravimetrically using a Cahn D200 recording ultra-microbalance with a mass resolution of  $\pm 0.1$  µg.

Controlled relative humidity was obtained by mixing dry gas and water saturated carrier gas streams (monitored by mass flow controllers), and measured continuously by a dew point controller. The temperature in the sample and the reference chambers was controlled to  $\pm 0.1$ °C by a water bath from Polyscience.

A sample size between 4 and 10 mg was used. Prior to being exposed to any water vapor, the samples were dried at 0%RH to remove any surface water present and to establish a dry, baseline mass. Next, the samples were exposed to four different relative humidity values: 20, 60, 75 and 80%RH. At each stage, the sample mass was allowed to reach equilibrium before the relative humidity was increased or decreased (the criterion for equilibrium was established when the weight variation did not exceed the value of 0,001 wt% during 5 min). If the equilibrium state was not reached after 1000 min, the relative humidity was changed to the next level automatically. The sorption isotherms were recorded at three different temperatures: 25, 40 and 50°C.

#### **Statistical Methods for the Predictive Modelling**

Analysis of data was performed using Enterprise Guide 4.3 with the SAS v9.2 software with PC Windows XP.

For the analysis, the crystallization onset  $t_{oc}$  is converted to log10.

## Preliminary Descriptive Analysis

For each %RH storage condition of interest (except for 0%RH), a linear regression was performed.

#### Data Analysis

An Analysis of Covariance (ANCOVA) was used to test the whether the variances for the data from the different % RH storage conditions could be pooled.

- If the test for equality of the slopes resulted in a *p*<0.25, then each condition was treated separately.
- If the test for equality of the slopes resulted in a p > 0.25 and the test for equality of the intercepts in a p < 0.25, the calculations were done for each condition but the common slope was used.
- If the test for equality of the slopes and intercepts both resulted in a p>0.25, the data were pooled.
- Predictive model: A 95% predictive confidence interval was obtained for the crystallization onset for samples stored for less than 3 months. The residuals were checked to verify that they were randomly distributed around 0 and fell between -2 and 2. The 95% one-sided lower confidence is used in order to predict the worst case scenario.

#### RESULTS

Amorphous solid dispersions of SAR/HPMCP (20/80) obtained by spray-drying, were stressed at different temperatures and humidities (Table II) for several months. At defined intervals, samples were analyzed by XRPD at room temperature to evaluate the time of onset of SAR crystallization. The impact of humidity on the  $T_g$  of the amorphous dispersion is evaluated before discussing the correlation between the crystallization onset and  $T_g/T$ .

## The Impact of Water Sorption on T<sub>g</sub>

The first step was the determination of the water sorption by the amorphous material. In Fig. 3 the weight variation is plotted as a function of %RH for three temperatures (25, 40 and 50°C). Interestingly, the three isotherms are superimposable. This can be explained by the fact that the heat of water desorption from the solid is close to the heat of pure water evaporation. It is usually the case for the hydration of organic compounds as shown by Khankari et al. (28). Both the equilibrium vapour pressure above the solid at given moisture, and that of pure water have a nearly parallel evolution with temperature according to Van't Hoff's equation, and the sorption isotherm expressed as a function of the %RH is nearly temperature independent. We did not measure the sorption isotherm at 60°C, for practical reasons, but based on the previous discussion we assumed that it was not significantly different from those obtained at the other temperatures.

The  $T_g$  of the spray-dried samples was measured by mDSC (using hermetically sealed pans) after equilibration at each %RH and at room temperature (23°C). There was no deliquescence at 100% RH. An example of the mDSC reversing curve is presented in Fig. 4. We observed a single  $T_g$  independently of the sorbed water amount suggesting that the water does not result in amorphous phase separation. The  $T_g$  can then be represented as a function of the



**Fig. 3** Water sorption measured by VTI for the SAR/HPMCP sample as a function of RH% at the different storage temperatures: 25°C, 40°C and 50°C.

**Fig. 4** mDSC "reversible" signal and  $T_g$  measurement for the SAR/HPMCP sample exposed to 80%RH, 23°C.



corresponding relative humidity. In Fig. 5, when the  $T_g$  of the solid dispersions is plotted as a function of the water content, a straight line is obtained which fits the theoretical Gordon-Taylor equation (Eq. 1) (29–31).

$$T_{g}^{GT} = \frac{w_{dry} T_{g}^{dry} + K w_{H_{2}O} T_{g}^{H_{2}O}}{w_{dry} + K w_{H_{2}O}}$$
(1)

where  $w_{dy}$  and  $w_{H_2O}$  are respectively the weight fraction of dry amorphous sample and of the adsorbed water on the total weight of the sample.  $T_g^{dy}$  is the  $T_g$  of the dried sample, and  $T_g^{H_2O}$  the  $T_g$  of pure water (135 K). *K* is a constant that can be approximated by the Sihma Boyer rule (29–31):

$$K = \left(\rho_{dy} \ T_g^{dy}\right) / \left(\rho_{H_2 0} \ T_g^{H_2 0}\right) \tag{2}$$



**Fig. 5**  $T_g$  as a function of water uptake. The symbols plot the  $T_g$  measured by MDSC, and the solid line represents the theoretical  $T_g$  according to Gordon-Taylor relation.

where  $\rho_{dy}$  and  $\rho_{H_20}$  are respectively the densities of the dry amorphous solid dispersion (estimated to be 1.31 g/cm<sup>3</sup>) and of the amorphous ice (of 0.9 g/cm<sup>3</sup>). The resulting value of *K* is 4.19.

The linear plot in Fig. 5 indicates that the plasticizing effect of water can be modelled by the Gordon Taylor equation.

# **Onset of Crystallization Time Measurements**

#### Results for $T < T_g$

The onset of the SAR's crystallization was monitored by XRPD at regular intervals of time. As an example, Fig. 6 illustrates the evolution of the XRPD diagram of the SAR sample stored at 23°C and 100%RH, from 3 to 6 months. For this particular example the first time where crystallization was detected after storage was 120 days ( $t_c$ ), and the last time where the sample was considered as purely amorphous was 90 days ( $t_{nc}$ ). Thus for this example the crystallization onset  $t_{ac}$  is defined as 105 days.

For each humidity and temperature storage conditions,  $T_g/T$ ,  $t_{nc}$ ,  $t_c$  and  $t_{oc}$  measured values are presented in Table III. The measurements for samples stored at 50°C and 75%RH were repeated twice, to test the reproducibility of this method. For practical reasons the stability study was limited to 17 months, and consequently the  $T_g/T$  values studied here do not exceed 1.15. In the following chapters we propose to extend the observed correlation law above this value, but we are aware that this assumption is not verified here.

#### Results for $T \ge T_g$

These analyses were performed directly in the temperature chamber of the X-ray powder diffractometer. The relative **Fig. 6** XRPD diagrams of the solid dispersion SAR/HPMCP stored at 23°C and 100%RH, at different times.



humidity was maintained at 0%RH by a nitrogen flow and the temperature was precisely monitored. The sample was left in the chamber at a fixed temperature above the  $T_g$ value (116°C) for several hours until crystallization occurred. As an example, some of the XRPD diagrams recorded at 126°C are shown in Fig. 7. These four diffractograms correspond to the following successive times of exposure: the initial time t0, the exposure time of 6.3 h corresponding to  $t_{nc}$ , the 7.4 h point corresponding to  $t_c$ , and the final time. From this data, the onset of crystallization at 126°C is about 6.8 h. All crystallization times are summarized in Table IV.

# Correlation between $T_g/T$ and $t_{oc}$

In Fig. 8 the crystallization time is plotted as a function of 1000/T in an Arrhenius plot for various %RH conditions. The vertical error bars are taken as the maximal uncertainty on the crystallization time measurements for each domain,

 $T < T_g$  and  $T > T_g$ . The horizontal error bars correspond to the uncertainty in temperature, of about 2°C for low temperature region ( $T < T_g$ ), and 0,5°C for high temperature region. It was estimated by comparing the set point of the oven regulation, or of the XRPD chamber, with local temperature measurements using calibrated probes. Figure 8 shows that in the low temperature domain  $T < T_g$ , the points are aligned on a different line for each humidity condition.

In Fig. 9,  $t_{oc}$  is plotted as a function of the rescaled reverse temperature  $T_g/T$ . The principal observations drawn from Fig. 9 are the following:

- There is a significant slope change around  $T_g/T=1$ , showing dramatically different behaviours on both sides of the  $T_g$ , as expected for  $\alpha$ -relaxation driven mechanisms.
- At  $T < T_g$  all crystallization onsets are well aligned *versus*  $T_g/T$  in the half logarithmic plot. The agreement is very good for the observation period of 17 months. This graph shows that the impact of both parameters,

Storage T (°C)	Storage RH (%)	%w/ w water	T <sub>g</sub> (°C)	T <sub>g</sub> /T	t <sub>nc</sub> (days)	t <sub>c</sub> (days)	t <sub>oc</sub> (days)	uncertainty%**
96	0	0%	116	1.054	30	34	32	6
60	80	5.6%	65	1.015	10	12	11	9
55	80	5.6%	65	1.031	16	19	17.5	9
50	100	nd	53	1.009	13	4	13.5	4
50	80	5.6%	65	1.046	25	31	28	11
50	75	4.9%	68	1.056	41	49	45	9
50	75	4.9%	68	1.056	46	53	49.5	7
40	100	nd	53	1.042	18	24	21	14
40	80	5.6%	65	1.080	65	72	68.5	5
40	60	3.7%	81	1.131	457	Nd (*)	Nd (*)	Nd (*)
23	100	nd	53	1.101	90	120	105	14
23	80	5.6%	65	1.142	300	330	315	5
23	75	4.9%	68	1.152	492	515	503.5	2

**Table III**Crystallization OnsetMeasurements for all the SamplesStored in Desiccators (at ControlledT and %RH) for T Inferiors to T<sub>g</sub>

(\*)The sample stored at 40°C and 60%RH had not yet crystallized after 457 days. This point is not plotted on the following graphs since technically there was no t<sub>m</sub> measured. but it is briefly mentioned in later discussion

(\*\*) the uncertainty is taken as  $(t_{nc} - t_c)/2 t_{oc}$ 



humidity and temperature of storage, can be described in terms of only one parameter,  $T_g/T$ , as was already observed by Miyazaki *et al.* (22) for the crystallization of nifedipine analogues; however in their study this was for  $T > T_g$ . The effect of the temperature rescaling is particularly dramatic for the observation of the point corresponding at 96°C, 0% RH ( $T_g/T=0.951$ ). In Fig. 8 this point is well separated from the other sub- $T_g$  points, whereas in the rescaled Fig. 9, it is well aligned with all other sub- $T_g$  points.

 At T > T<sub>g</sub>: there is a non Arrhenius behavior, which can correspond to the Adam-Gibbs-Vogel (AGV) model (see below), with much shorter crystallization times and a higher temperature dependence.

In the Appendix it is demonstrated by statistical means that there is no significant difference of behaviour for the three main humidity conditions: 75%RH, 80%RH and 100%RH.

#### DISCUSSION

#### **Theoretical Interpretation**

#### α–relaxations

In this section we propose an explanation of the observed results by the link between  $\alpha$ -relaxations and crystallization.

The  $\alpha$ -relaxations behave differently above and below the glass transition temperature, according to the Adam Gibbs model (6,14,32,33).

# T >Tg

The molecular  $\alpha$ -relaxation time  $\tau_{\alpha}$  is related to the temperature by the Vogel-Tammann-Fulcher (VTF) equation:

$$\tau_{\alpha} = \tau_{\alpha 0} \exp\left(\frac{D T_0}{T - T_0}\right) \tag{3}$$

where  $\tau_{\alpha \theta}$  is the limit for relaxation time at the high temperature limit, and  $T_{\theta}$  is the temperature where  $\tau$  becomes infinite.  $T_{\theta}$  is often taken as the Kauzman temperature and D is a material specific parameter depending on the glass fragility. D is related to the fragility index m (defined in Eq. 4) by Eq. 5.

$$m = \left[\frac{d\log(\tau_{\alpha}(T))}{d(T_g/T)}\right]_{T=T_g}$$
(4)

$$D = \frac{2.303 \ m_{\min}^2}{m - m_{\min}} \tag{5}$$

The minimum value of m,  $m_{min}$ , is close to 16, (34), thus:

$$m = m_{\min} + \frac{2.303 \ m_{\min}^2}{D} \approx 16 + \frac{590}{D} \tag{6}$$

Storage T (°C)	Storage RH (%)	T <sub>g</sub> (°C)	T <sub>g</sub> /T	t <sub>nc</sub> (days)	t <sub>c</sub> (days)	t <sub>oc</sub> (days)	Uncertainty (%)
115	0	116	1.003	2.52	2.69	2.61	3
116	0	116	1.000	0.77	0.88	0.83	7
126	0	116	0.975	0.26	0.31	0.28	8
136	0	116	0.951	0.05	0.07	0.06	19

2800



Fig. 8 Crystallization's onset as a function of 1000/T.

$$\frac{T_0}{T_g} = \left(1 - \frac{m_{\min}}{m}\right) \tag{7}$$

# $T < T_g$

The glassy state below  $T_g$  is a non-equilibrium one (33), where the macroscopic configurational entropy remains constant and depends only on the fictive temperature  $T_{f}$ As a consequence,  $\tau_{\alpha}$  has Arrhenius-like temperature dependence. The most common relation used to fit the temperature dependence to the relaxation times in this range is the so-called Adam-Gibbs-Vogel (AGV) equation (35):

$$\tau_{\alpha} = \tau_{\alpha 0} \exp\left(\frac{D T_0}{T \left(1 - T_0 / T_f\right)}\right) \tag{8}$$

Pikal and co-workers (21) have shown that it is possible to fit experimental measurements of  $\tau$  above and below  $T_g$ (measured by dielectric spectroscopy above  $T_g$  and by



**Fig. 9** Crystallization's onset as a function of  $T_{g}/T_{c}$ 



Fig. 10 Experimental data versus modified VTF/AGV equations.

calorimetric method below  $T_{g}$  using the same set of values for D and  $T_{\theta}$ .

#### Crystallization Onset: *a*-relaxations Relationship

As mentioned in the introduction, several studies (13,20)have related the chemical degradation or the crystallization rate in amorphous systems to the different relaxation times. In order to correlate the onset crystallization time  $(\tau_{oc})$  to the  $\alpha$  relaxation time we propose to use an equation very close to that already proposed by Bughra et al. (13):

$$\log(\tau_{oc}) = \xi \, \log(\tau_{\alpha}) + \log(\tau_{oc}^{0}) \tag{9}$$

where  $\xi \leq 1$  is a coupling parameter and  $\tau_{oc}^{0}$  is a constant.

When  $T > T_g$ , one expects therefore an apparent VTF law for crystallization time, as follows

$$\log(\tau_{oc}) = \log(\tau_{oc}^{0}) + \frac{1}{\ln(10)} \times \frac{D'T_{0}}{T - T_{0}}$$
(10)

By fitting experimental values of  $\tau_{oc}$  one may obtain  $\tau_{oc}^{0}$ ,  $D' = \zeta D$  and  $T_{\theta}$ .

At  $T < T_{\varrho}$ , the above equation may be modified according to AGV theory to:

$$\log(\tau_{oc}) = \log(\tau_{oc}^{0}) + \frac{1}{\ln(10)} \times \frac{\xi D T_{0}}{T(1 - T_{0}/T_{f})}$$
(11)

Table ∨ Fitting Parame- ters for the VTF Derived Equations (Eq. 8)	<i>T</i> ₀c <sup>°</sup>	7.95 10 <sup>-7</sup> days
	Т <sub>о</sub> D'	324 К (Т <sub>g</sub> -Т <sub>0</sub> =65 К) 2.99
	D ξ	96 7.4 0.40

Table VI	VTF Parameters	Values from	Shamblin et a	1. 1999	(37)
	1 1 1 4 4 4 1 0 0 0 1 0	101000 11 0111	01100111010111000		

	T <sub>g</sub> (K)	Т <sub>0</sub> (К)	Т <sub>g</sub> - Т <sub>о</sub> (К)	D
Sorbitol	264	224	40	7.8
Sucrose	348	290	58	7.3
Trehalose	378	322	56	5.I
Indomethacin	322	256	66	8.9

# Humidity Effect

When a hygroscopic amorphous solid is exposed to humidity,  $T_g$  usually rapidly decreases. However it has been shown that D is mostly independent of the moisture (27,36).

Therefore one can assume that, while  $T_0$ ,  $T_g$ ,  $T_f$  change will moisture uptake, the ratios  $T_0/T_g$  will remain constant, according to Eq. 7, and so will do the  $T_0/T_f$  since  $T_f$  and  $T_g$  are very close. The Eq. 11 can be so rewritten:

$$\ln(\tau_{oc}) = \ln(\tau_{oc}^{0}) + \frac{\xi D T_{0}/T_{g}}{(T/T_{g}) \times (1 - T_{0}/T_{f})}$$
(12)

$$\ln(\tau_{oc}(T, RH)) = \log(\tau_{oc}^{0}) + A \ \frac{T_g(RH)}{T}$$
(13)

where  $\boldsymbol{A}$  is a constant:

$$A = \frac{\xi D T_0 / T_g}{(1 - T_0 / T_f)}$$
(14)

Therefore the plot of  $\tau_{oc}$  versus  $T_{p}/T$  is expected to be linear.

# **Comparison to the Experimental Results**

In Fig. 10 we have superimposed the experimental results and the theoretical results according to the modified VFT/AGV equations.

#### At $T \ge T_g$ (0% RH)

Table VII Parameter Estimates

With 3 Months Data

The curve has been fitted to the experimental data by using the empirical equations quoted above (essentially the Eqs. 6 and 10). Best fit parameters ( $\tau_{oc}^{\circ}$ ,  $T_0$  and D') are given in Table V. Knowing that  $T_g$  at 0% RH was measured as 389 K (116°C), we can obtain: the fragility index *m* from the  $T_0/T_g$  ratio (Eq. 7), the *D* parameter from the VFT



**Fig. 11** Linear regression with 3 month data. \* on this figure, the  $t_{oc}$  is noted as tmc, the mean crystallization time.

<3 months (used for regression fit)

1.15

Tq/T

1.20

>3 months

95% prediction

1.25

1.30

equation, and finally the  $\xi$  coupling exponent (Eq. 9). The *m* value obtained corresponds to that for a fragile glass.  $T_g$ - $T_o$  (= 65 K) and *D* values are in the range of values previously reported by Shamblin and collaborators (37) (cf Table VI). The  $\xi$  value is lower than those reported by Bhugra and Pikal (38) for pure products (0.75 for nifedipine and 1 for phenobarbital).

# At $T \leq T_g$ (Various RH)

log10(tmc)

1.00

1.05

0

1.10

Regression fit

At  $T < T_g$  we traced the theoretical onset of crystallization according to Eq. 11, by using the  $\tau_{oc}^{\circ}$ ,  $T_0$  and D' values obtained for the  $T > T_g$  domain. Two hypotheses were evaluated:

- first a classical assumption that T<sub>f</sub>∼ T<sub>g</sub> (corresponding to the full red line),
- secondly that  $T_{f}/T_{g}=0.98$  whatever the moisture content (corresponding to the dotted red line). Note that  $T_{f}/T_{g}=$  0.98 corresponds to  $T_{f}=T_{g}$ -8 K at 0% RH. This second point thus presumes that the actual product underwent some aging during its production process.

It is very clear from the plot that the crystallization times predicted for  $T_f = T_g$  (no aging of the glass) are about one order of magnitude lower than the experimental data, even for Tvery close to  $T_g$ . In contrast, the prediction with  $T_f/T_g \sim 0.98$  is much closer to the experimental data. In particular it explains a jump of the crystallization time between experiments just

Label	DF	Parameter Estimate	StandardError	t Value	Pr >  t	95% Confid	ence Limits
Intercept		- 10.75361	1.35152	-7.96	000.>	-13.94946	-7.55777
Tg/T		11.68012	1.29538	9.02	000.>	8.61703	14.74321

Storage T (°C)	Storage RH (%)	T <sub>g</sub> /T	Experimental t <sub>oc</sub> (days)	Predicted t <sub>oc</sub> (days)	Calculated t <sub>m</sub> (days) 95% lower limit	Calculated t <sub>m</sub> (days) 95% upper limit	In the 95% prediction interval
23	100	1.101	105	130	83	199	Yes (-19%)
40	60	1.131	>570	287	150	555	No (>+100%)
23	80	1.142	315	385	185	814	Yes (-16%)
23	75	1.152	504	506	224	1151	Yes (-0.4%)

Table VIII Extrapolated mean crystallization time calculated from Eq. 14 and compared to experimental values

above and just below  $T_{g}$ . However we may observe some deviation of the slope of the experimental correlation *versus* the theoretical line. This may be the consequence of a slight aging when the  $T_g/T$  values increase.

Finally, even if the theory - experiment agreement is not perfect, these data confirm that the presence of moisture leads to decreases in  $T_g$ ,  $T_{\theta}$ , and  $T_f$  while keeping mostly constant the  $T_g/T_{\theta}$  and  $T_f/T_0$  ratios. In addition, moisture does not change the fragility index *m*. The data also suggest that an experiment performed at  $T > T_g$  cannot predict the crystallization behaviour at  $T < T_g$ , if one does not take into account the history of the glassy material (as expressed by  $T_f$ ). Therefore one may expect a different crystallization rate for samples of the same composition obtained by a different process or even significantly different process conditions.

# Long-Term Stability Prediction from 3 Month Data

In this section we examine the data corresponding to  $T < T_g$  from an empirical and pragmatic point of view and determine if it is possible to predict the long term stability (up to 17 months) based on mid-term stability data (up to 3 month) according to rigorous statistical analysis (Table VII).

We fitted a linear model of Log  $10(t_{ac})$  versus  $T_g/T$ . From the fitted model with data of less than 3 months, the



Fig. 12 Inverse prediction (model with the 3 months data).

predicted mean response with a 95% predictive confidence interval is illustrated in Fig. 11. One can note that data above 3 months fall within the 95% prediction interval and are very close to the fit line.

In Table VIII, each one of the four experimental points is individually compared to the theoretical extrapolated values and to the limits of the 95% prediction interval. The experimental values are relatively close to the calculated ones, excepted for the point corresponding to the conditions 40°C and 60%RH. This point is not represented on the graphs since the sample had not yet crystallized after 17 months, but it would be outside the 95% prediction interval. This exceptional result might be an artefact or might be due to an unexplained phenomenon occurring during the storage. One hypothesis is that when an amorphous sample is stored for such a long time in such warm conditions, its real density increases significantly, and the molecular mobility is reduced.

# **Design Space**

The prediction can also be used to design storage conditions from mid-term stability data. The inverse prediction, giving the  $T_{g}/T$  values from crystallization times (see Fig. 12), is presented in Table IX for some classical stability times (1 to 6 months; 1, 2 and 4 years). For example, the usual shelf life for pharmaceutical drugs is 2 years (730 days). The minimum  $T_{g}/T$  value to reach at least 730 days without crystallization should be more than 1.17 according to the regression line, or 1.20 when taking a safety margin at a 95% lower statistical limit.

Table IX         Inverse Pre-           diction at Different Days	Days	T <sub>g</sub> /T (95% one-sided lower confidence bound)		
	30	1.05 (1.05)		
	90	1.09 (1.10)		
<sup>a</sup> Prodiction at 730 and	180	1.11 (1.13)		
1095 days are out of range. these results must be used/interpreted with caution	365	1.14 (1.17)		
	730	1.17 (1.20)		
	1095	1.18 (1.22)		





Fig. 13 The design space giving the storage conditions for different target shelf-lives.

By taking into account the correlation that was found between  $T_g$  and RH, a design space can be defined in the temperature and RH space for 1, 2 or 4 years. The design space presented in Fig. 13 is based on the lower statistical limit, by taking into account the safety margin. This graph predicts that, with a safety margin of 95%, this amorphous material could recrystallize between 1 and 2 years when stored at 25°C and 60% RH. It would therefore be necessary to store it under specific conditions, for example either 5°C or with desiccants at ambient temperature, in order to minimize any risk of crystallization. A 40°C temperature should be totally avoided.

# CONCLUSION

In this study, crystallization times were measured for a pharmaceutical amorphous dispersion stored at different temperatures T below  $T_g$ , and different relative humidities. Results indicate that this crystallization time is correlated very simply to only one parameter,  $T_g/T$ . Moreover it is demonstrated that for  $T < T_g$ , the relation between the crystallization's onset and the rescaled inverse temperature  $T_g/T$  is an exponential law strictly independent of the %*RH*.

Fig. 14 Linear regression of the  $t_{mc}$  (or  $t_{oc}$ ) as a function of  $T_gT$  for different humidities (75, 80 and 100%RH).

This can be explained by noting that the moisture content does not change the D parameter, and therefore maintains a constant ratio of  $T_g/T_0$ . Below  $T_g$ , an effect of aging from the material's history is observed, meaning that the crystallization time is probably dependent on the material and the process to obtain it.

From a practical point of view, if the mid-term data ( $t_{oc}$  inferior to 3 months) are analyzed, and if the resulting exponential law is extrapolated to 18 months, it has been demonstrated that no significant discrepancy is observed between the calculated crystallization times and the measured ones. Consequently, this methodology can be used to design accelerated stability studies in order to anticipate physical destabilization for an amorphous drug product. The ultimate result of such an approach would be the definition of long term shelf life as early as possible in pharmaceutical development (even before phase 1 clinical studies), and the early design of appropriate packaging, adapted to the desired shelf life.

Finally, some restricted extensions of this methodology might be applicable for stability prediction. The extrapolation law, measured on a specific batch, might be extended to other batches as long as they are similar regarding certain properties. Indeed some physico-chemical characteristics seem to be very important for the kinetics of recrystallization. The residual solvent quantity can change the  $T_g$  of the drug product just like

 Table X
 Table of ANCOVA: Type 3 Tests of Fixed Effects

Effect	Num DF	Den DF	F Value	$\Pr > F$
H0: Equality of the slopes	2	5	1.54	0.3015
H'0: Equality of adjusted means	2	5	1.45	0.3181

H0: since p > 0.25. there is sufficient evidence to consider a common slope

H'0: since p < 0.25. there is sufficient evidence to consider a common intercept

water. The thermal history of the batches, as determined by the primary drying in the spray-drier and the secondary drying can anneal the amorphous powder and affect its molecular mobility. All such properties, which would potentially influence the drug product stability, could be identified as critical quality attributes of the amorphous intermediate drug product in a Quality by Design approach.

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# **APPENDIX: STATISTICAL CALCULATIONS**

#### **Data Pooling**

The first question was: can we consider that the data obtained at 75, 80 and 100% RH is a unique population? A preliminary Analysis Of Covariance (ANCOVA) was conducted to test whether the data of the different %RH storage conditions could be pooled.

The table of ANCOVA (Table X) shows that the points  $(T_g/T, \log 10(t_{mc}))$  for the different humidity conditions can be considered from a statistical point of view as the same population. The first important conclusion, is that the %RH impact should be taken into account only once in the calculation of  $T_g$ .

As illustration, we show the 3 different linear regressions for the different %RH on the same graph, see Fig. 14.

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